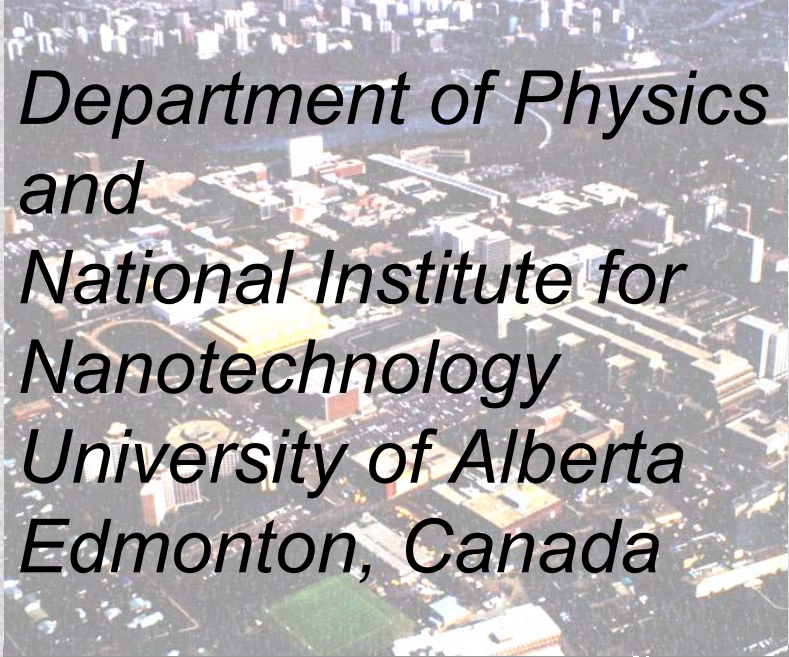


# Ultrafast magnetization dynamics: from continuous films to nanoparticles

Mark Freeman

with Miro Belov, Kristen Buchanan, BC Choi, Grey Arnup,  
Sasha Krichevsky, Zhigang Liu, Al Meldrum, Xiaobin Zhu



*Department of Physics  
and  
National Institute for  
Nanotechnology  
University of Alberta  
Edmonton, Canada*

Supported by  
NSERC  
iCORE  
CIAR  
CRC

MAG = 25.00 K X  
EHT = 8.00 kV

2  $\mu$ m

Stage X = 10.111 mm  
Stage Y = 14.145 mm  
Stage Z = 10.111 mm  
Stage Delta X = 0.001 mm  
Stage Delta Y = 0.001 mm  
Stage Delta Z = 0.001 mm

Detector

Time : 18

Stage Delta Z

Workshop on Nanomagnetism Using X-ray Techniques, 30 Aug 2004

**motivation:** study the influence of structure on magnetization dynamics

**approach:** time-domain, local probing, including stroboscopic imaging and “ripple tank” experiments with magnetic oscillations in mesoscopic ferromagnets

**theme for the workshop:** opportunities created by the ability to attack these problems with x-rays

- major gains: spatial resolution (including ability to image static configurations with high resolution, important as initial conditions for dynamics); chemical selectivity;
- neutral: temporal resolution
- sacrifices: capability for vector measurement; ability to probe stochastic phenomena, potential for single-shot imaging

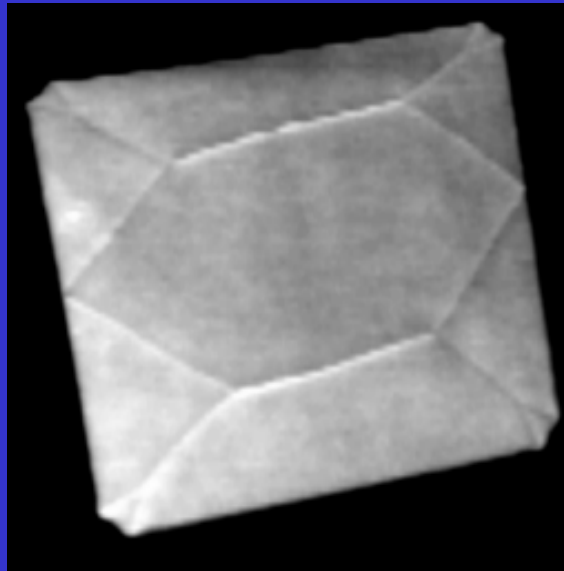
# outline

- mode control in lithographic elements: small angle precession
  - large angle excitation during switching:
    - oscillations and damping
  - nanocrystalline composites:
    - exchange and dipolar vs. only dipolar coupling between “giant classical spins”
- comparison of experiment and numerical modeling
  - discussion and prospects

# Magnetic Structures

Magnetic domain structure is a function of the geometry:

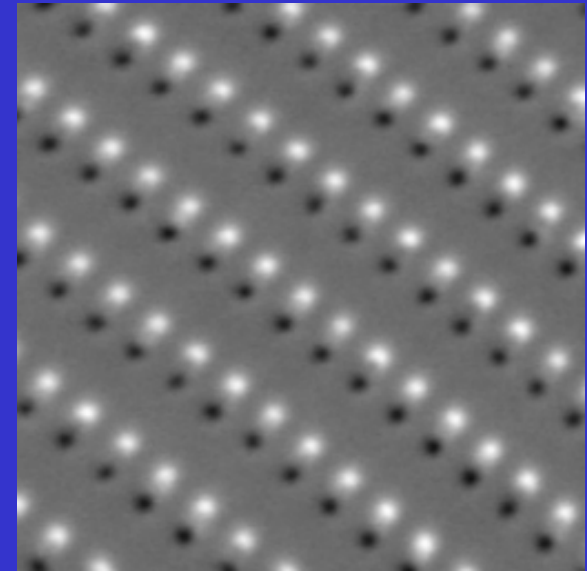
patterned film



25  $\mu\text{m}$  x 25  $\mu\text{m}$

20  $\mu\text{m}$  permalloy square

single domain

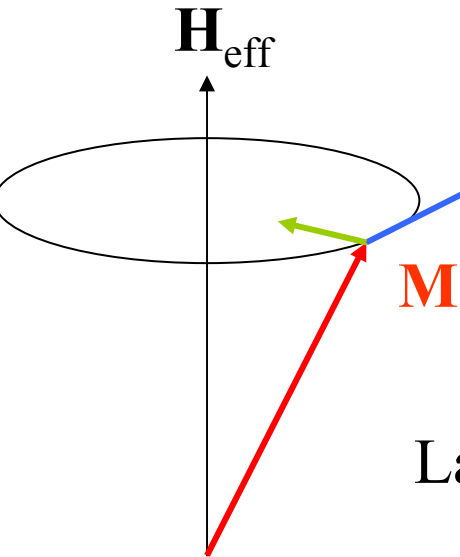


3  $\mu\text{m}$  x 3  $\mu\text{m}$

Particle size:  
240 nm x 90 nm x 10 nm

Xiaobin Zhu,  
MFM

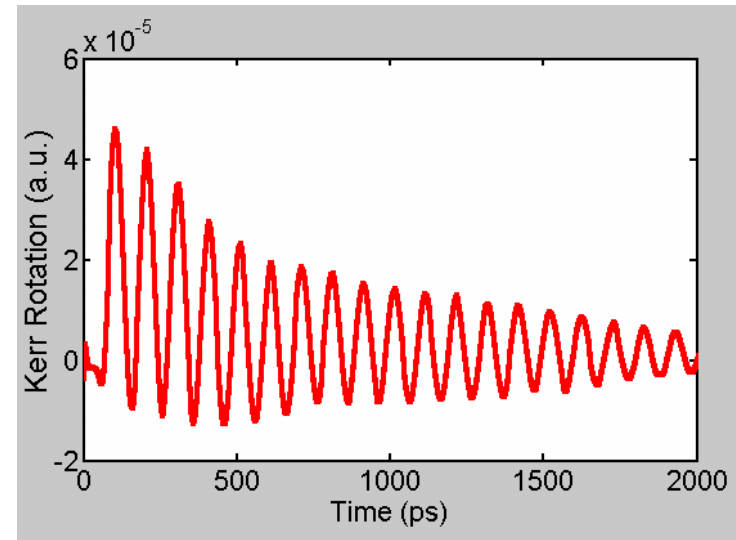
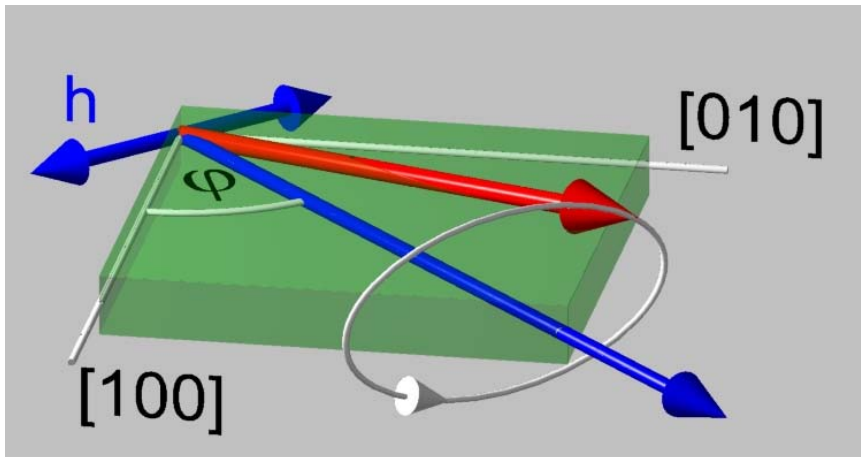
# Spin Dynamics



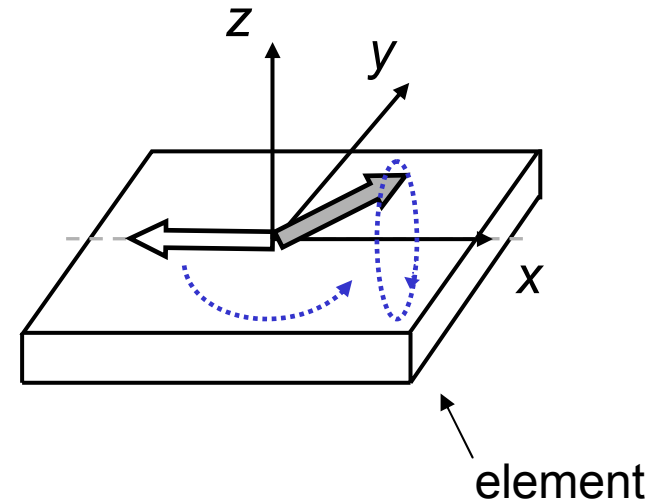
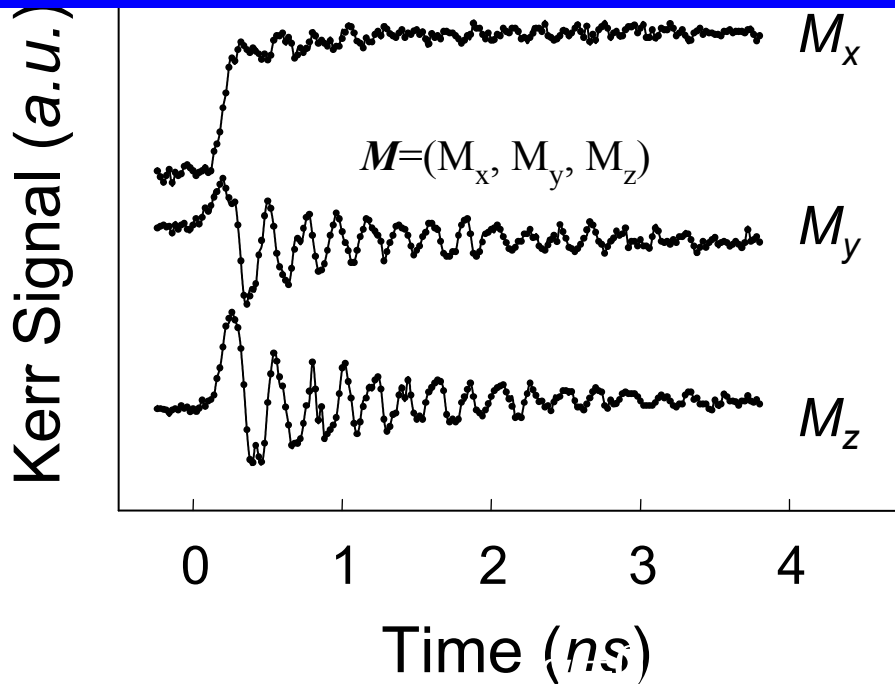
$$\frac{\partial \mathbf{M}}{\partial t} = \gamma [\mathbf{M} \times \mathbf{H}_{eff}] - \alpha \gamma [\mathbf{M} \times \frac{\partial \mathbf{M}}{\partial t} \times \mathbf{H}_{eff}]$$

Landau-Lifshitz-Gilbert Equation

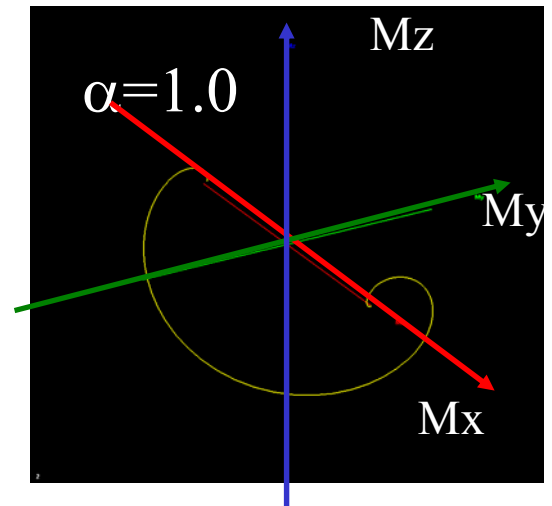
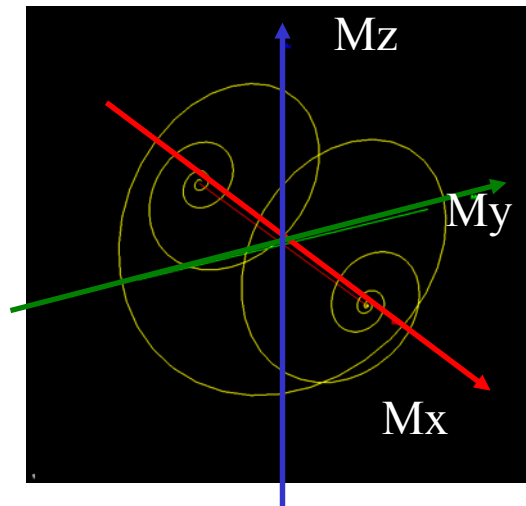
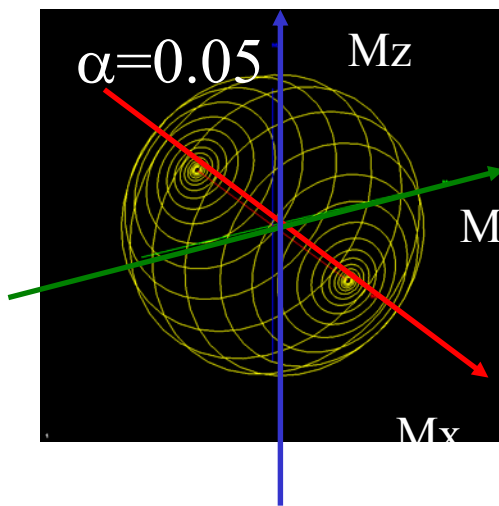
$\alpha$ : 0.002 (Fe)    0.011 (Co)  
0.019 (Ni)    0.006 (NiFe)



# Large Angle Precession, Switching

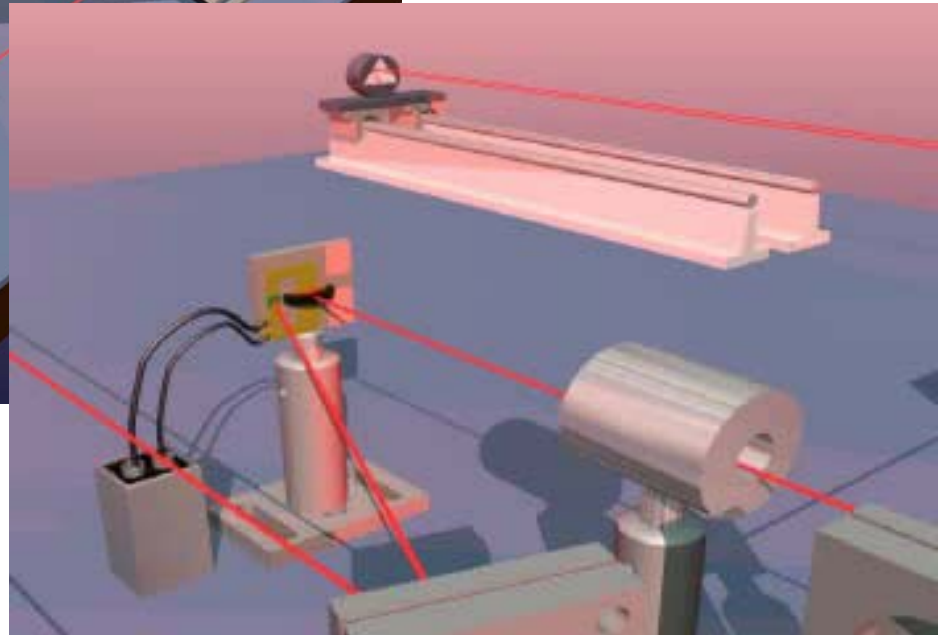
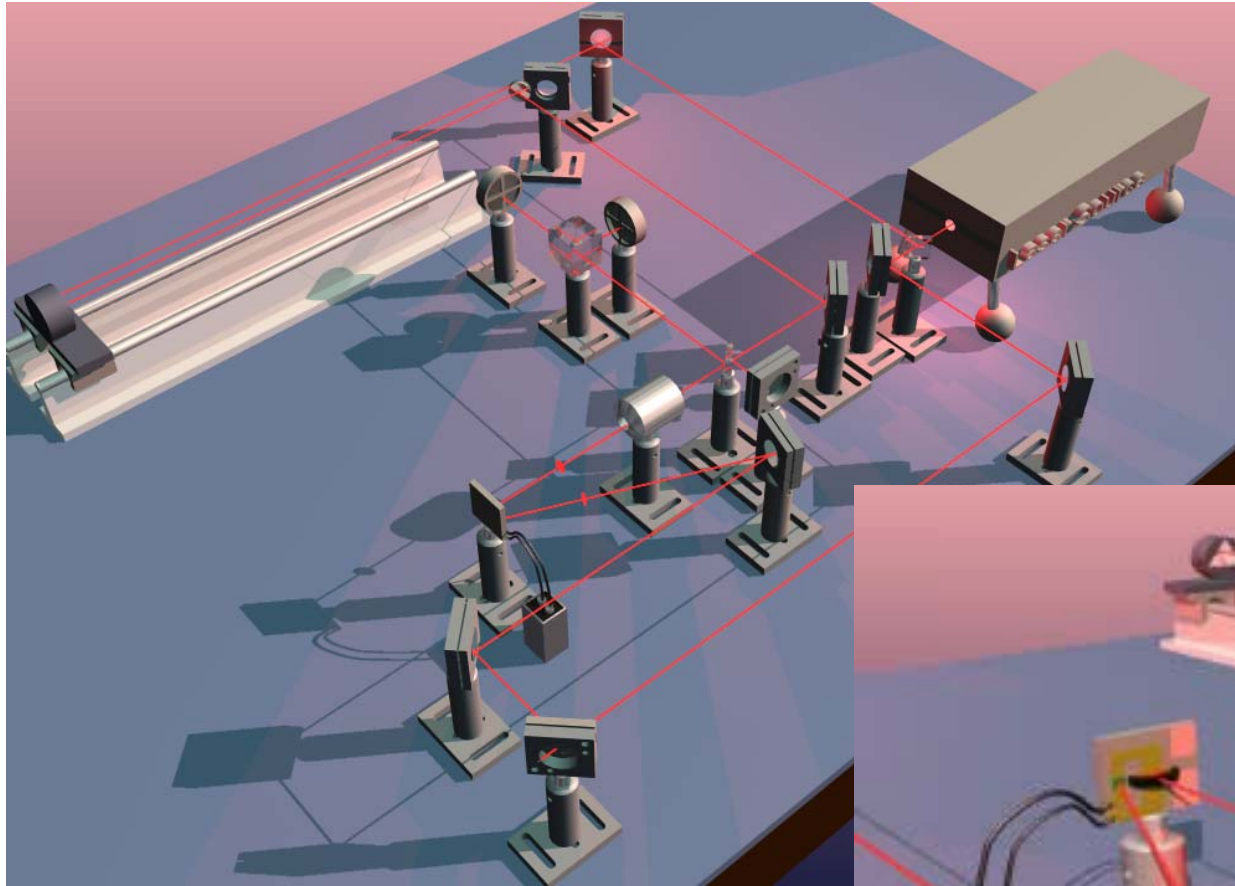


Oscillation after 'switching'



# time-resolved scanning Kerr microscopy

high bandwidth from short optical pulses; high-speed information  
from relative timing



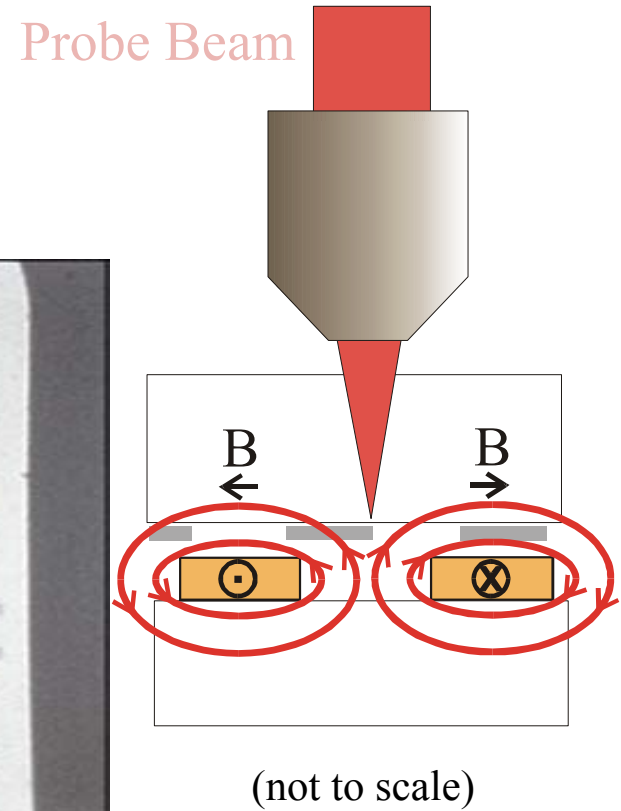
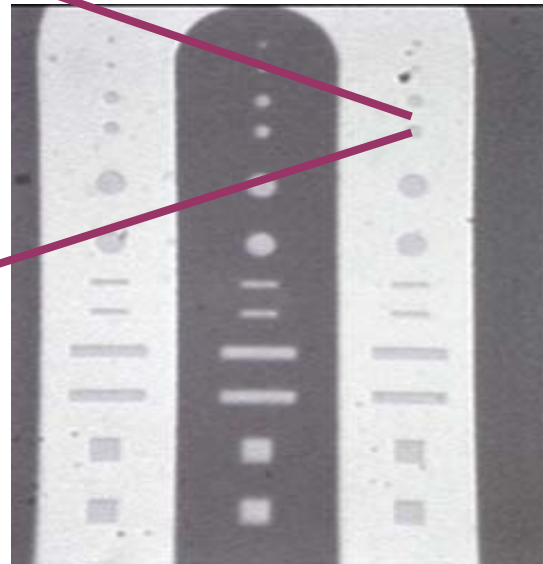
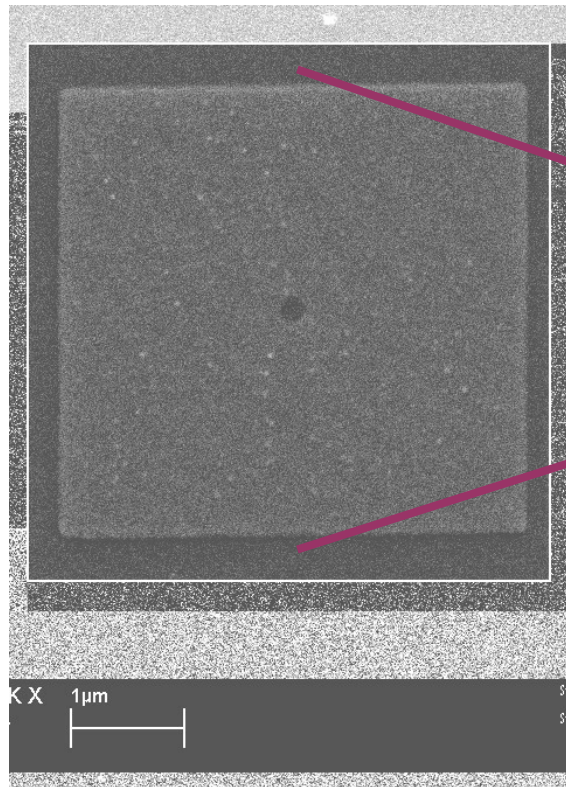
Dave Fortin

<http://nanoscale.phys.ualberta.ca>



# time-resolved scanning Kerr microscopy

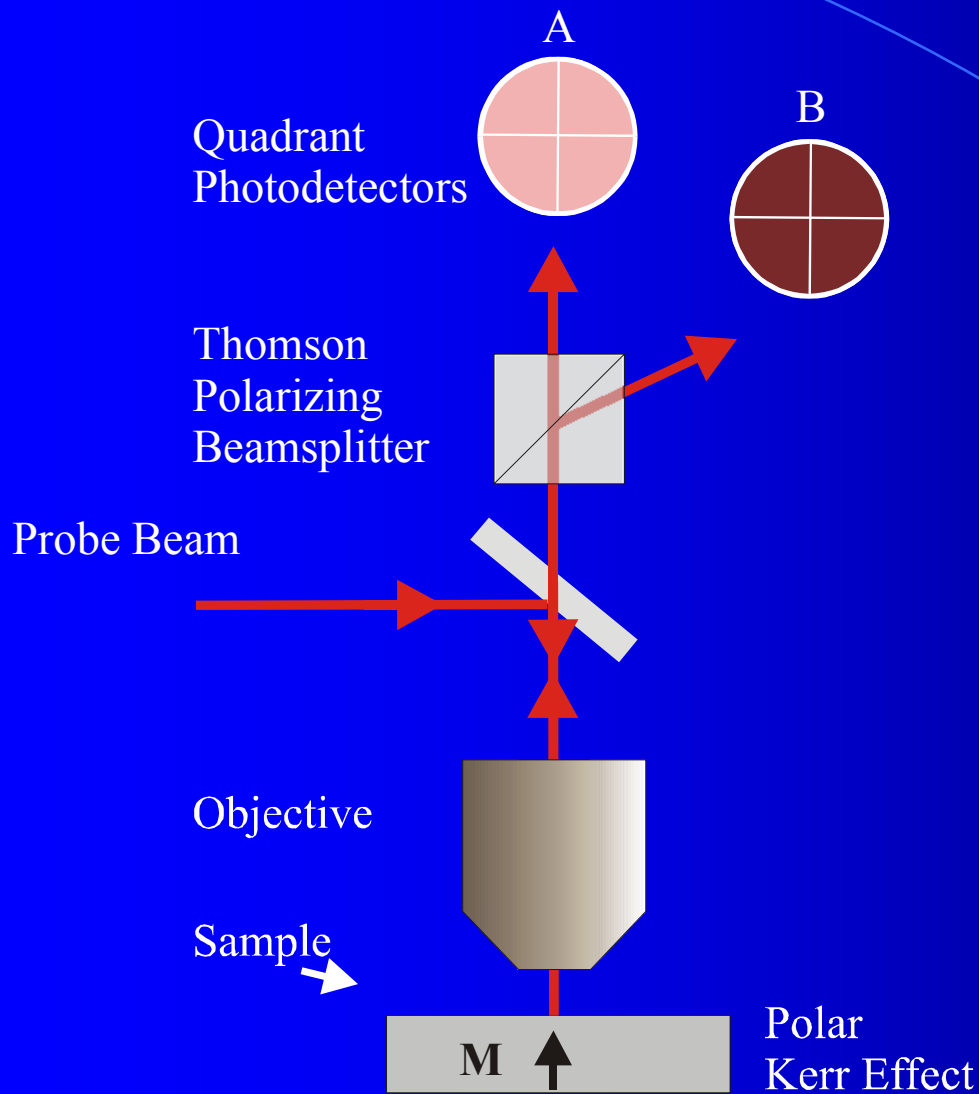
specimen geometry for both in- and out-of-plane transient excitation



- “pump” provides fast transient magnetic field (current pulse traveling through transmission line)

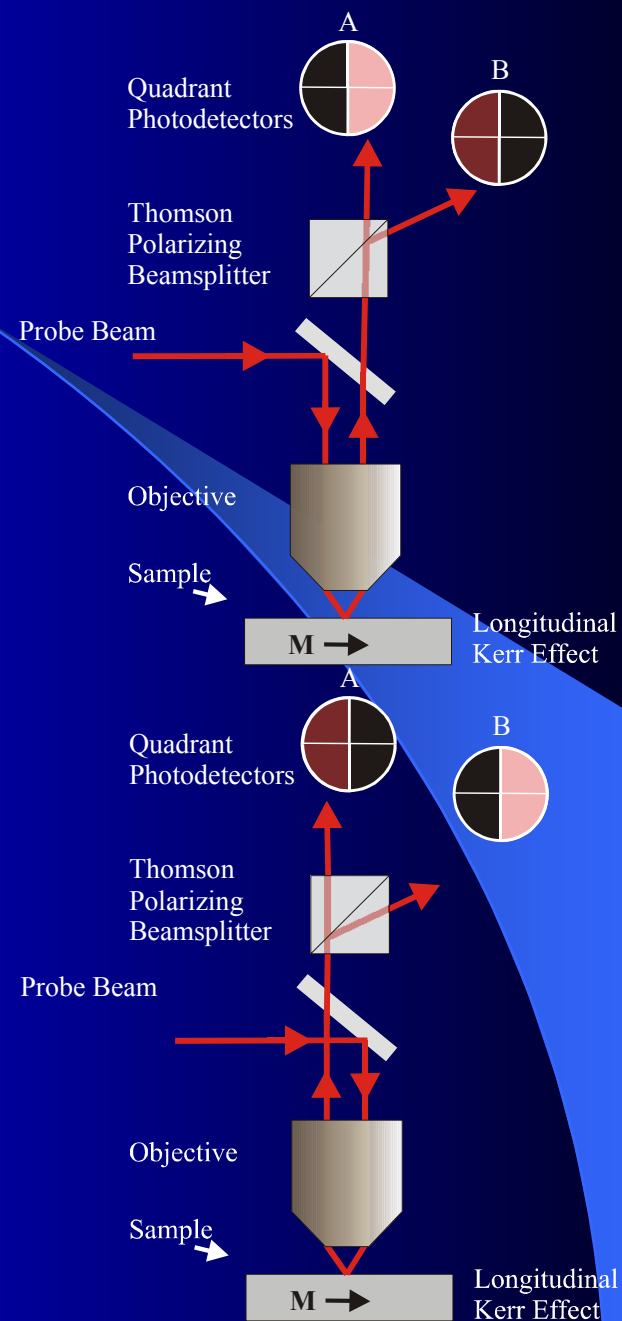


# "vector magnetometer": quadrant detection scheme



A. Krichevsky,  
K. Buchanan

polar

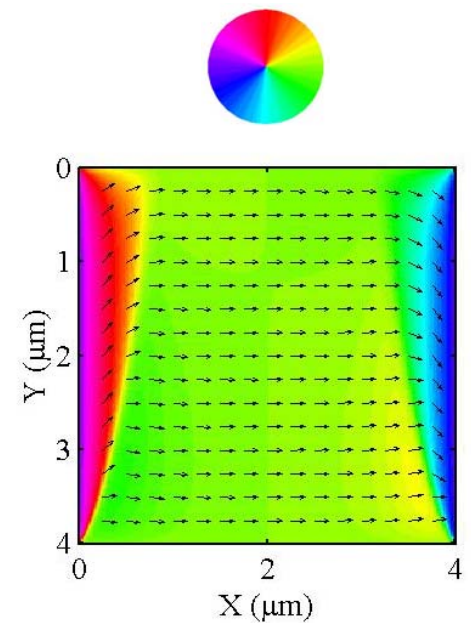
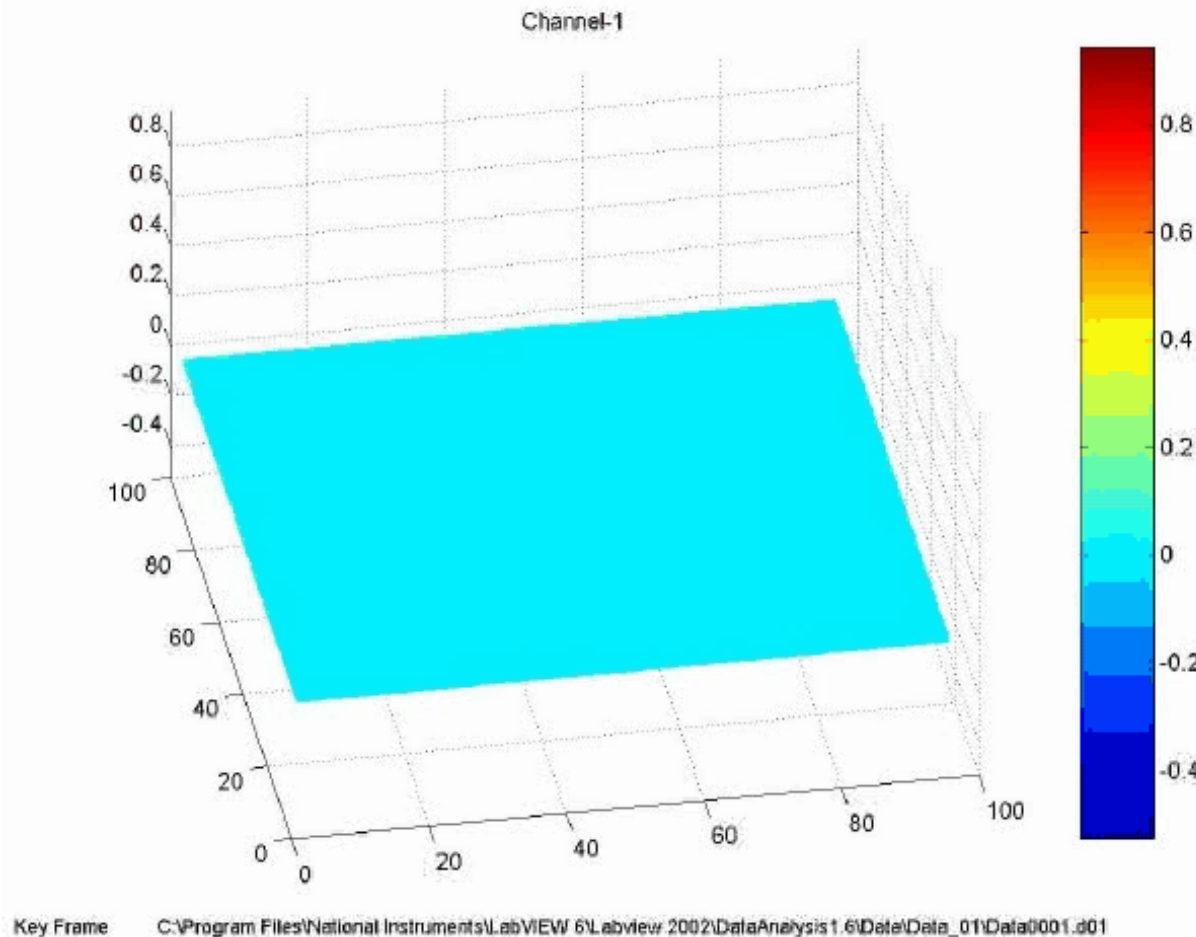


longitudinal

# FMR response to transient out-of-plane field

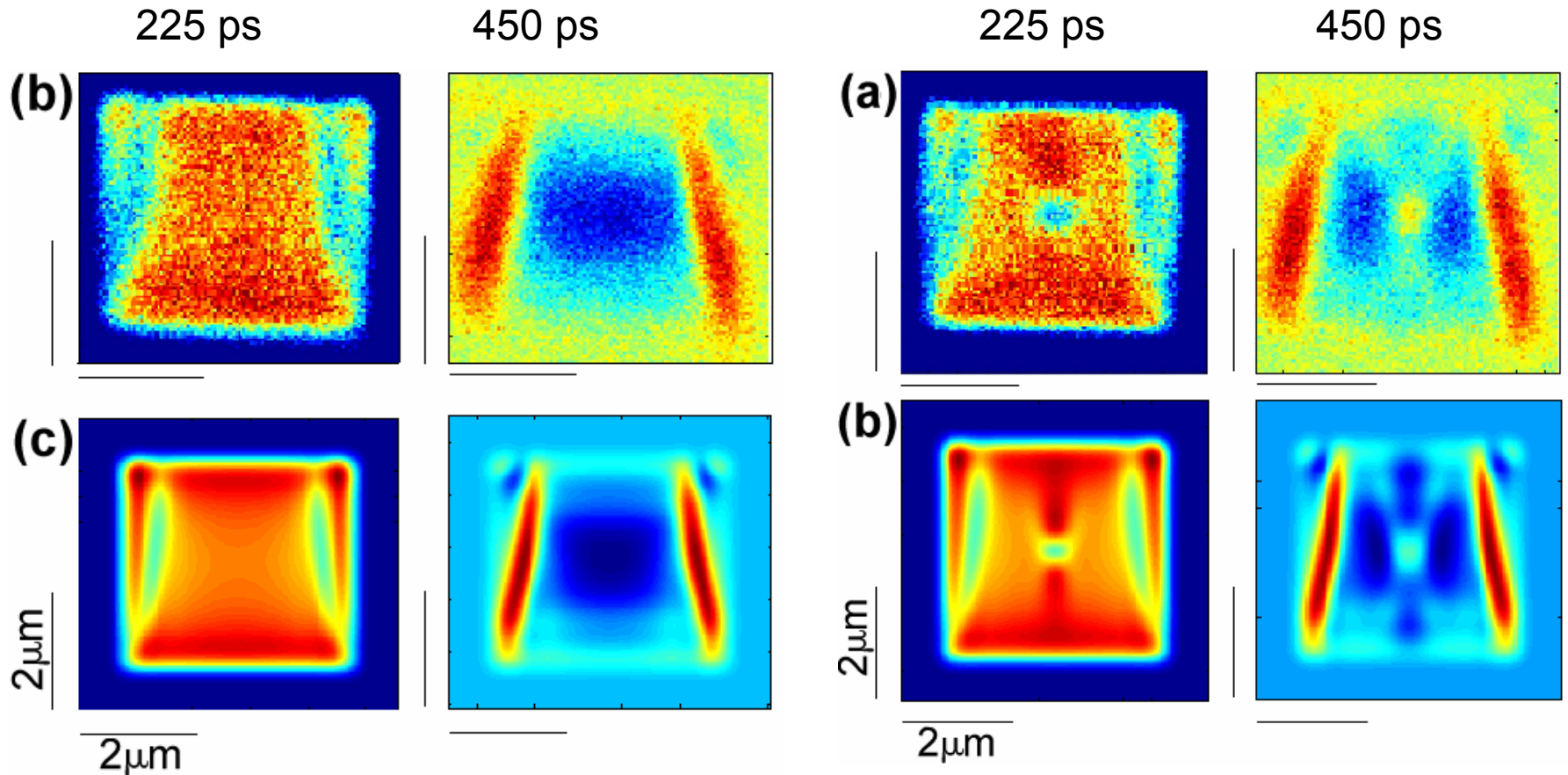
small tipping angle, 51 Oe DC bias field

polar Kerr signal:



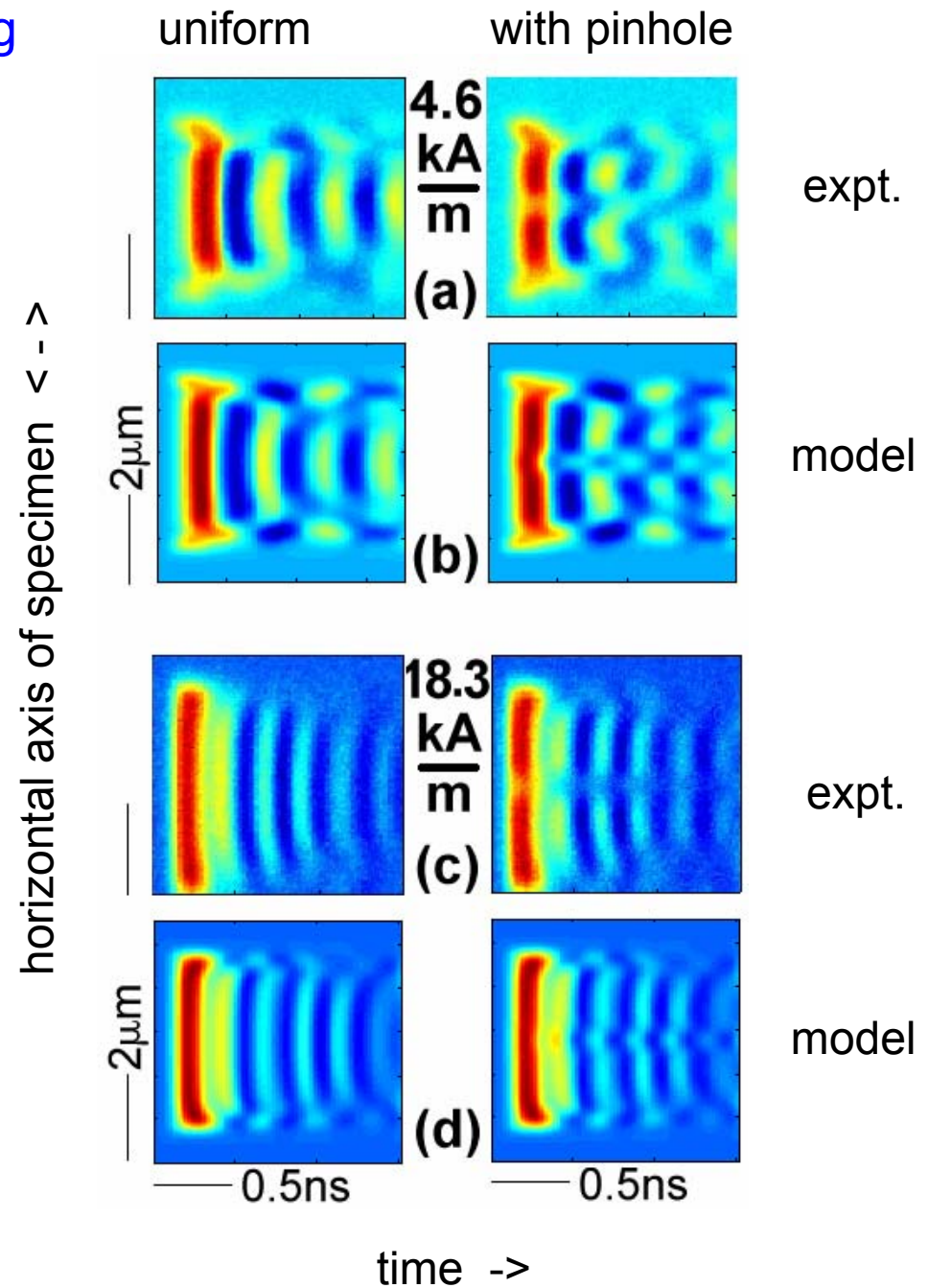
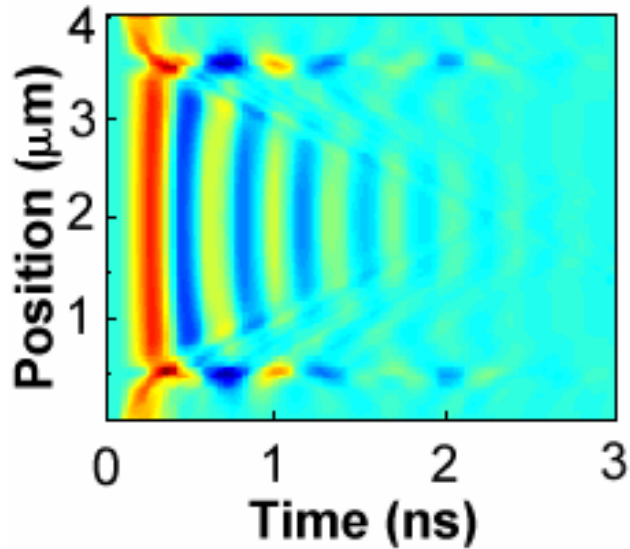
initial C state  
(from simulation)

“snapshots” of out-of-plane magnetization,  
square specimen with and without pinhole  
4.6 kA/m in-plane bias

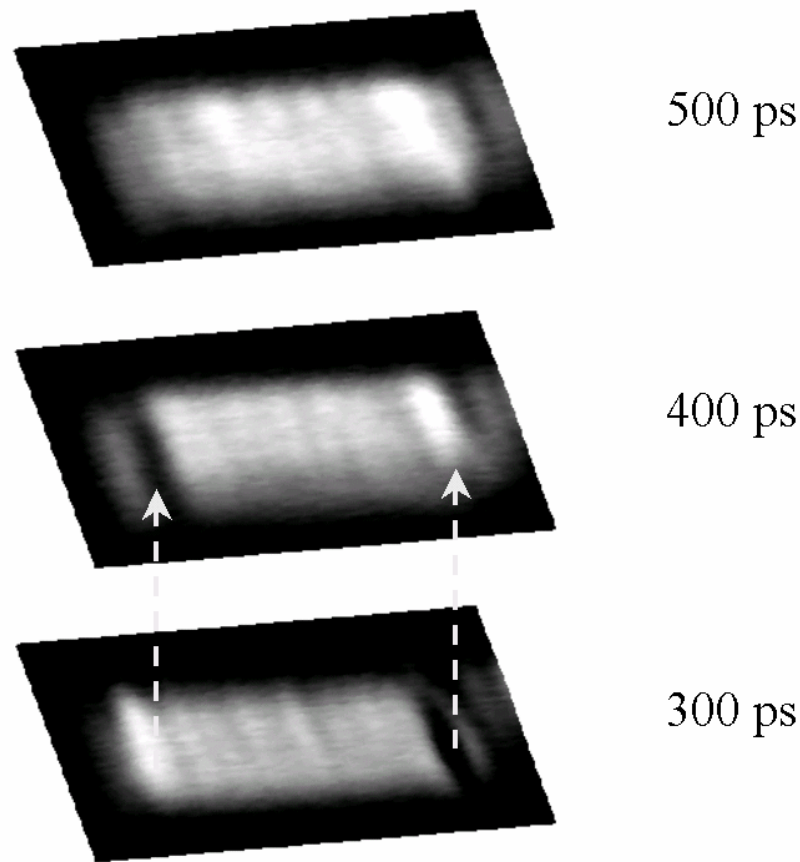
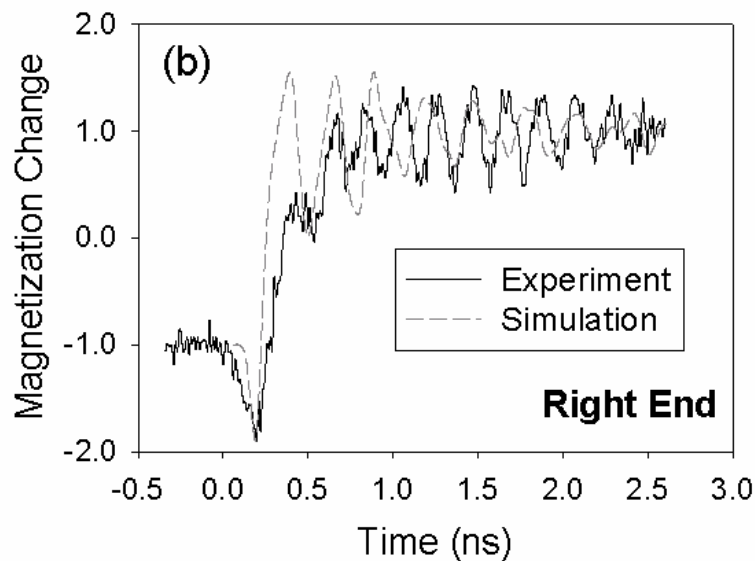
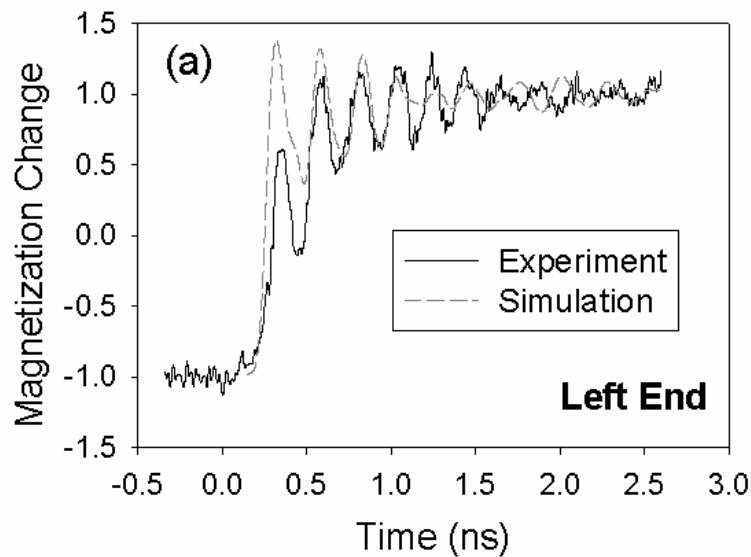


modal oscillations conform to symmetry of nonuniform magnetization

position-dependent effective damping  
as revealed in “x vs. t” cross sections

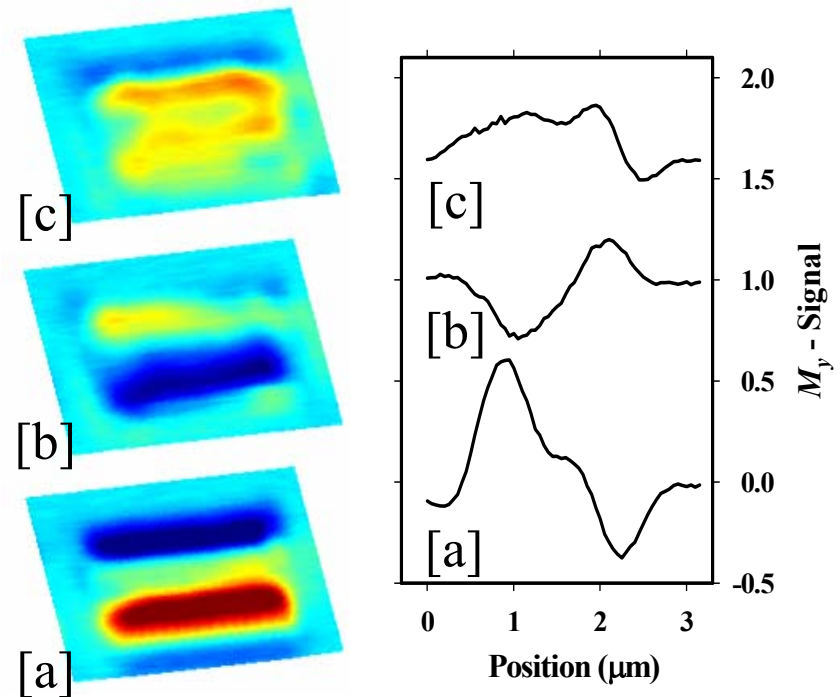
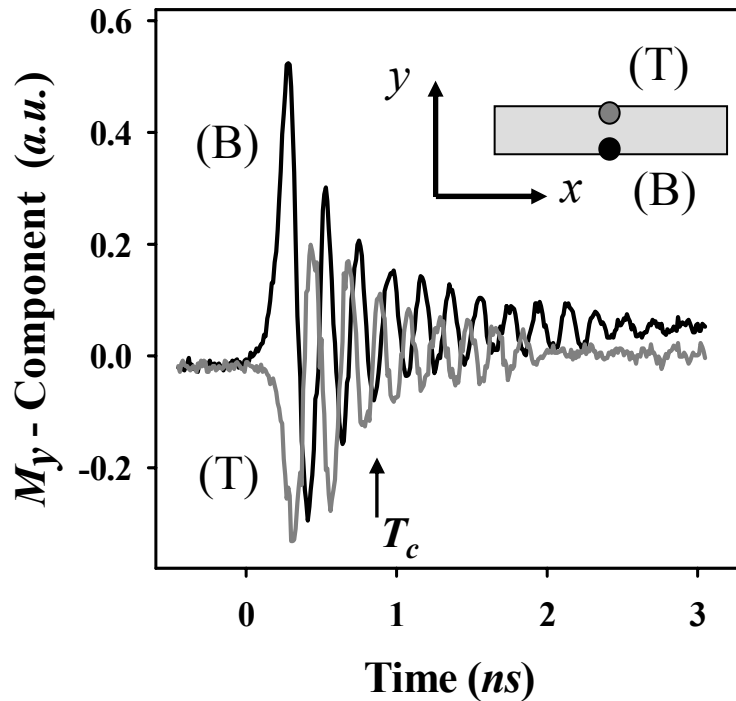


## large amplitude excitation of low frequency closure domain mode



- this measurement is an admixture of the longitudinal (easy-axis) and polar components
- perhaps stronger suppression of initial out-of-plane excursion in expt. relative to sim. owing to sample imperfections; finite spatial resolution of msmt.

transient closure domains also exhibit large amplitude oscillations  
form at long sides owing to dipolar interaction resisting switching of the edges

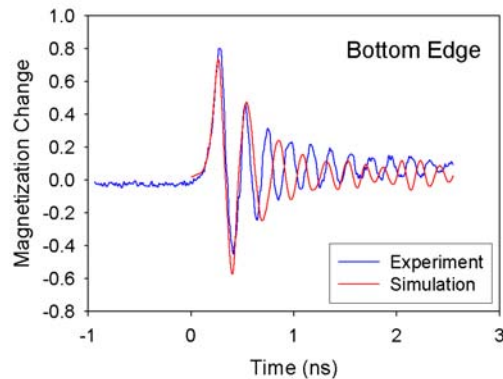
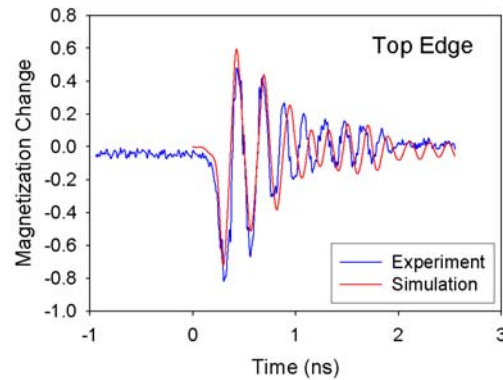


time-resolved images of in-plane  
hard-axis magnetization component,  
with line cuts

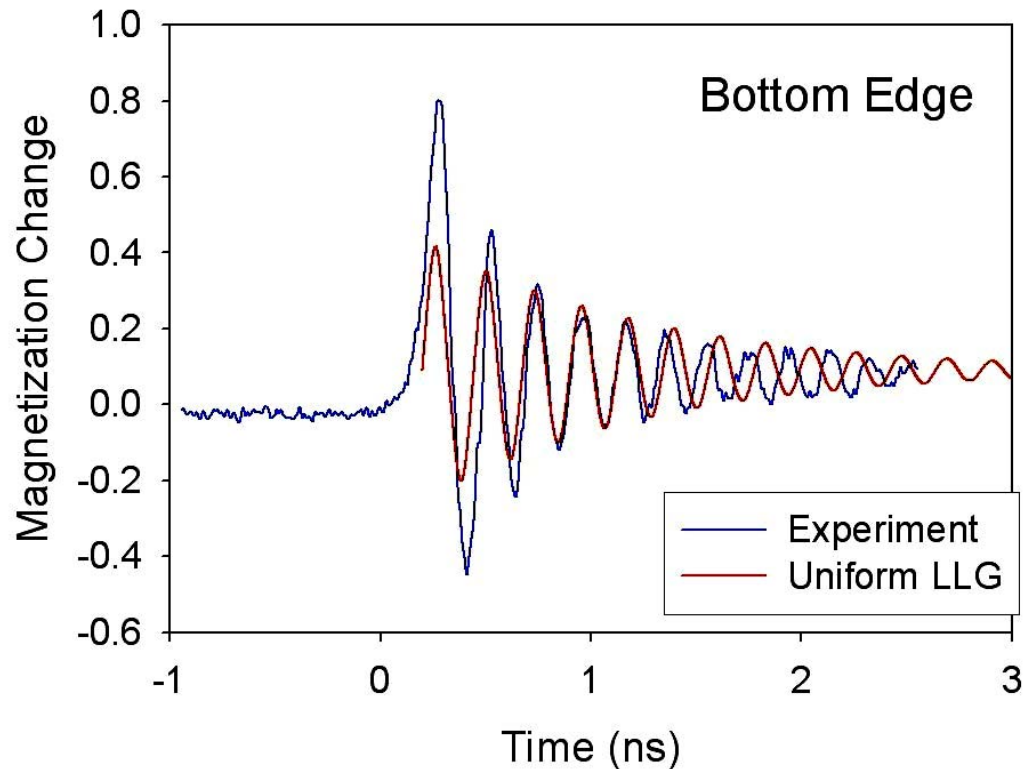


envelope of the decay:

accelerated early ringdown, owing to energy transfer to short wavelength modes;  
reasonably accounted for by finite-element LLG

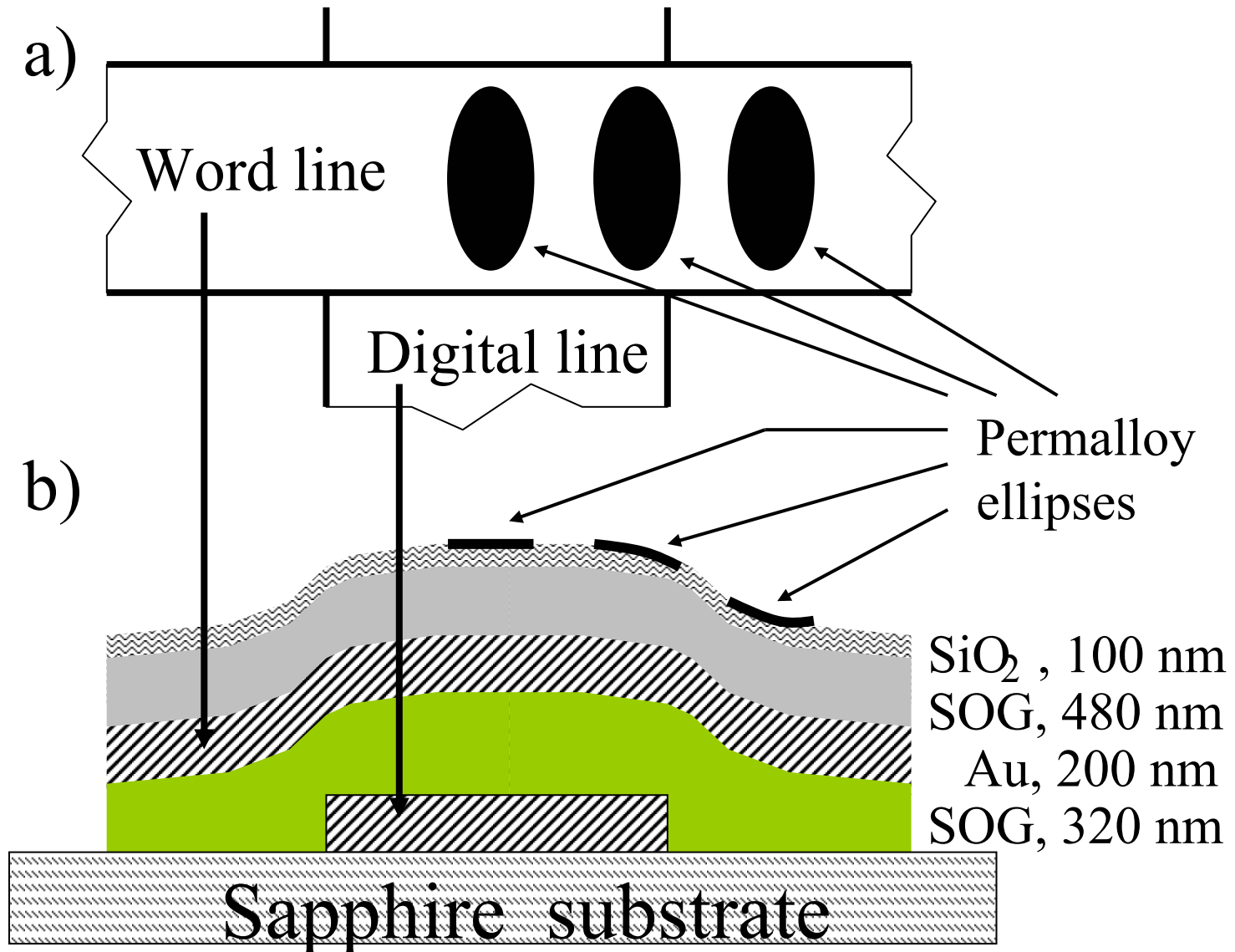


comparison to simulation:  
finite element LLG with  $\alpha = 0.01$



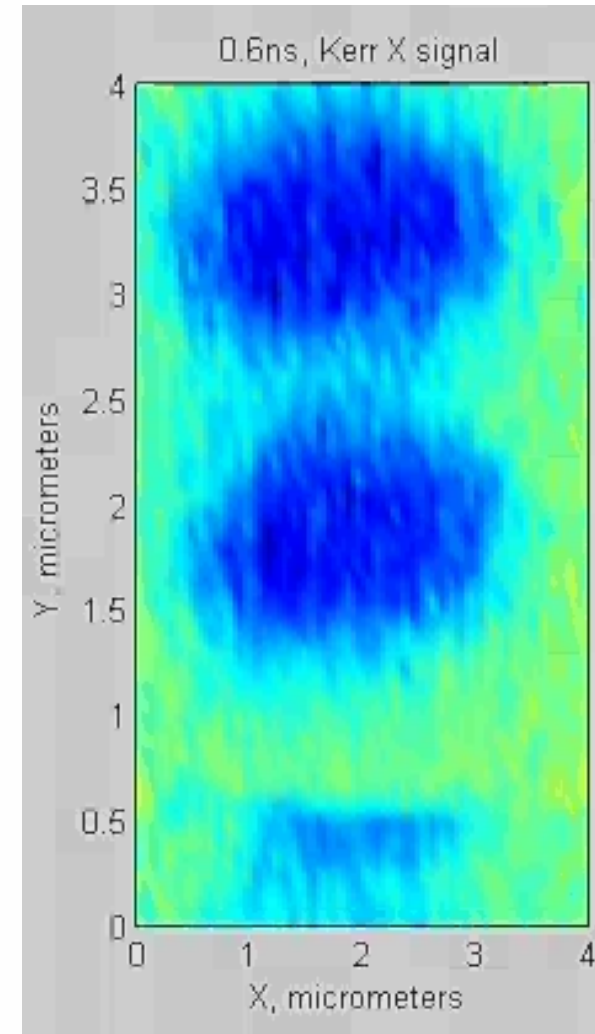
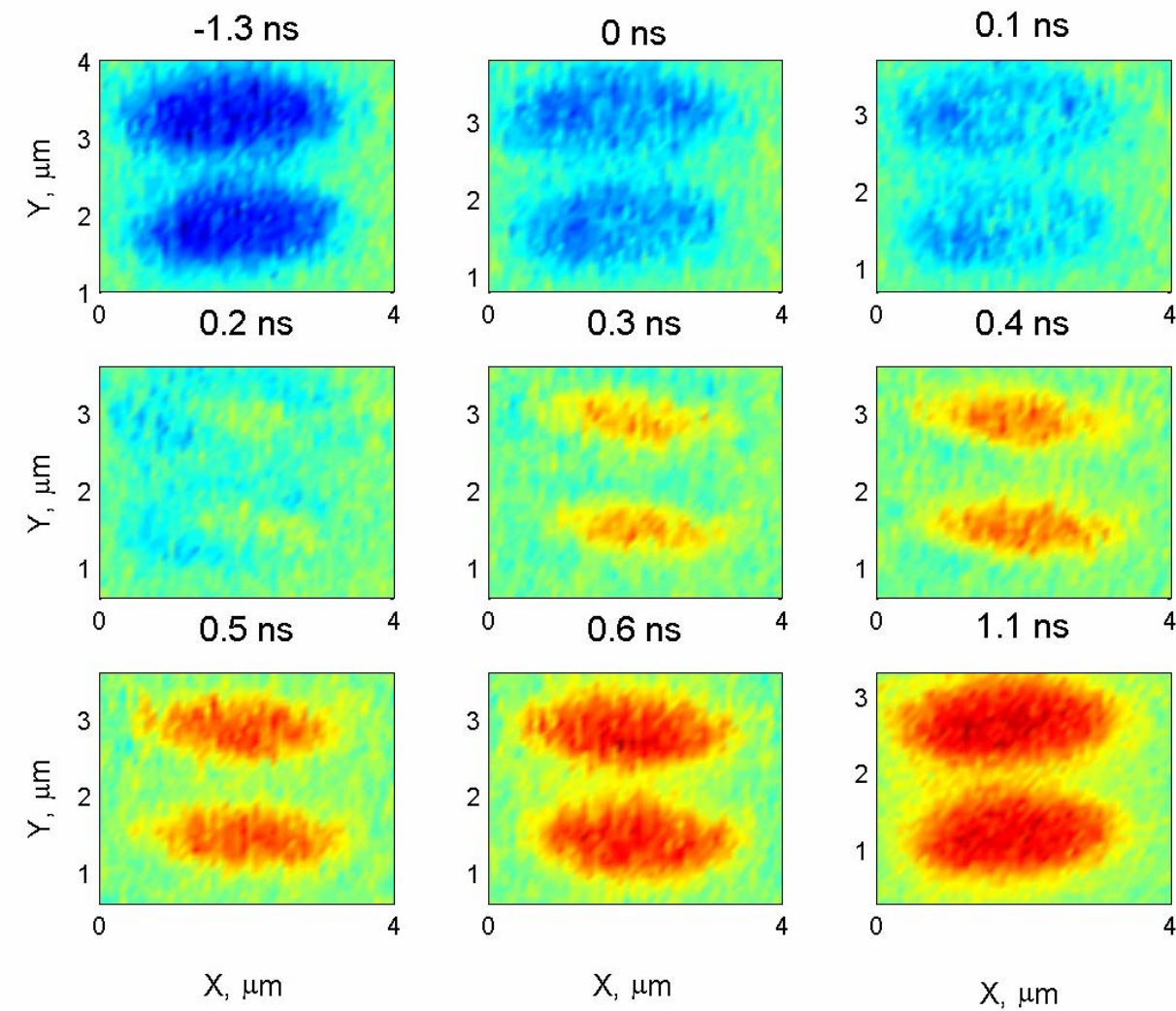
comparison to simulation:  
uniform LLG with  $\alpha = 0.01$

switching with simultaneously acting easy- and hard-axis pulses  
in typical MRAM geometry



example of switching scenario in “half-select” regime

3x1  $\mu\text{m}$  ellipses;  
movie duration – 2.6 ns.

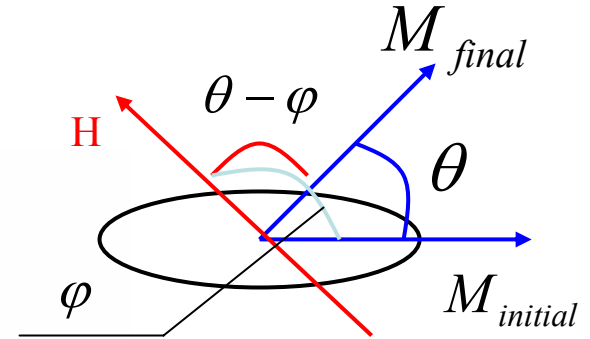
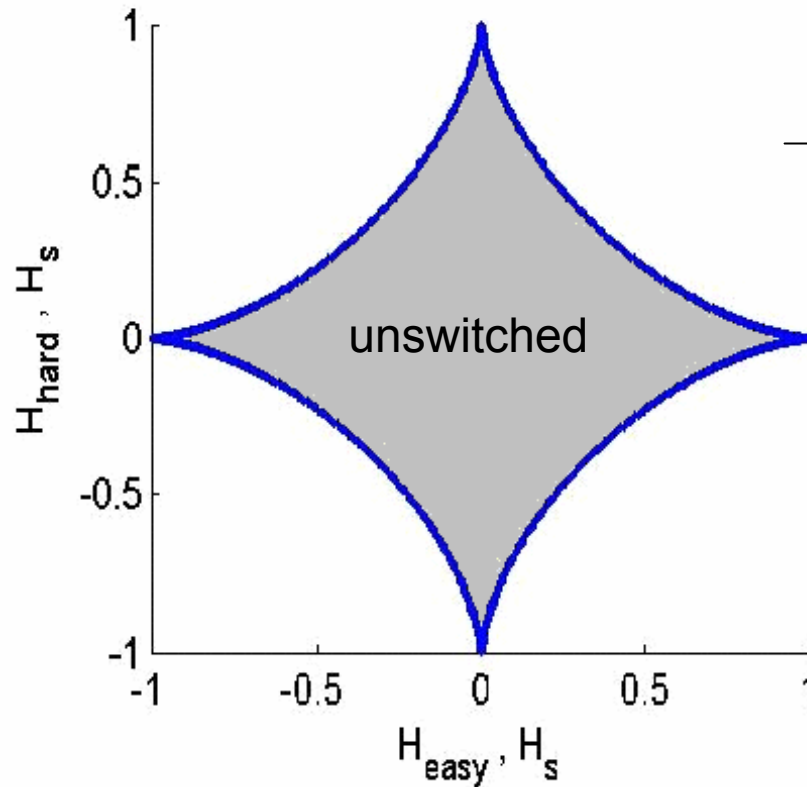


“half-select” switching: simultaneous action of easy- and hard-axis fields required to flip

### Stoner-Wolfarth astroid

Energy expression for uniaxial anisotropy:

$$E = K \sin^2 \theta - H\mu \cos(\varphi - \theta)$$



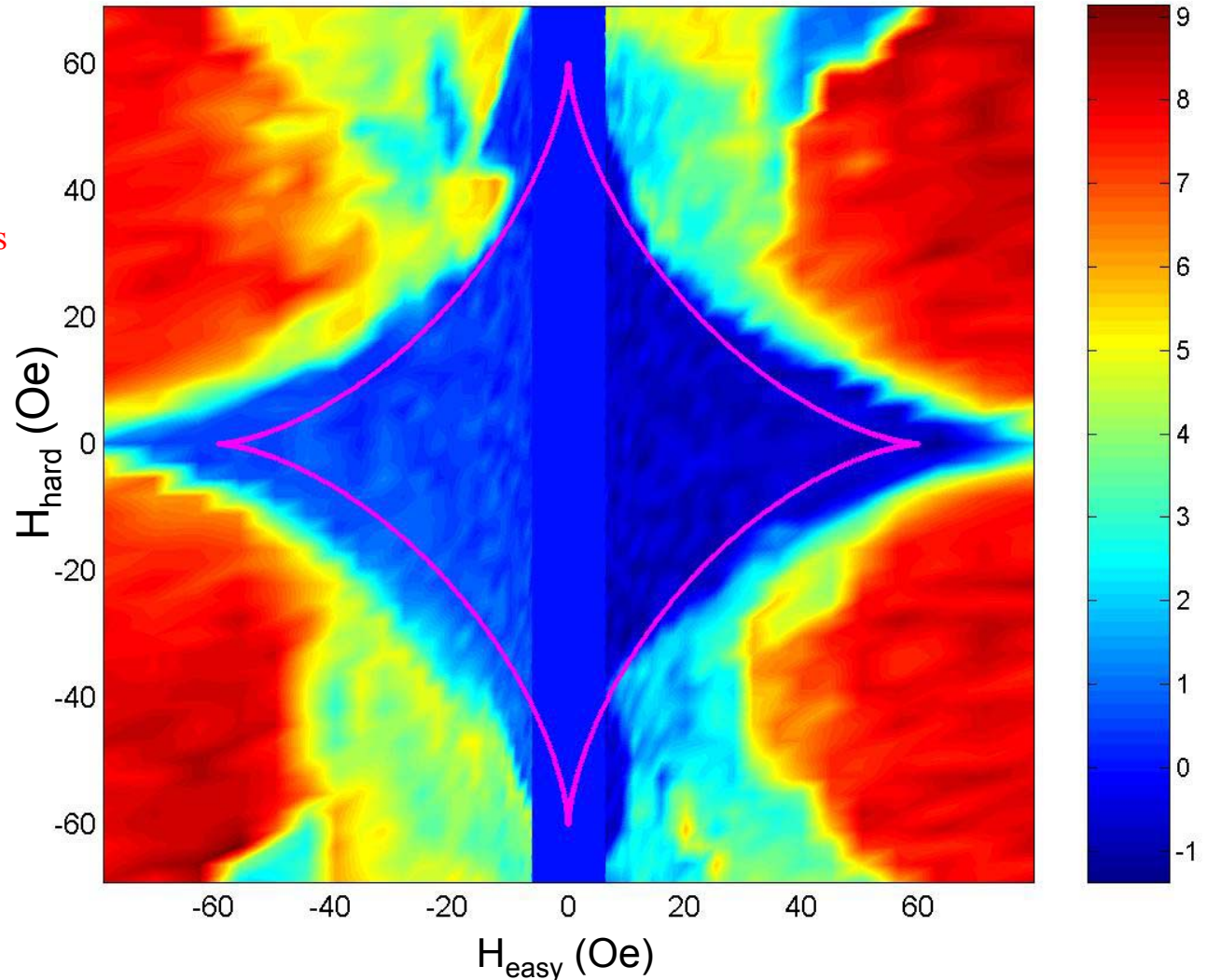
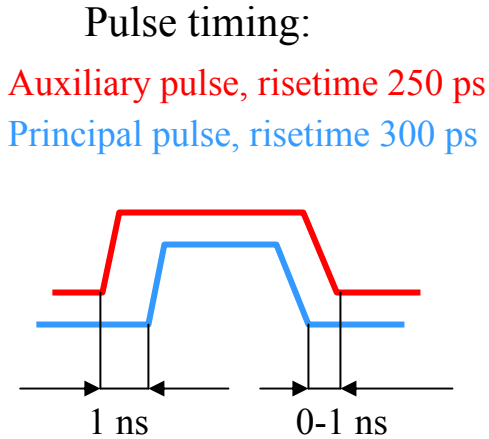
$$H_x^{2/3} + H_y^{2/3} = H_{\text{coercivity}}^{2/3}$$

# Switching diagram, easy-axis pulse width 0.5 ns

Final state magnetization is measured 20 ns after the pulses end.

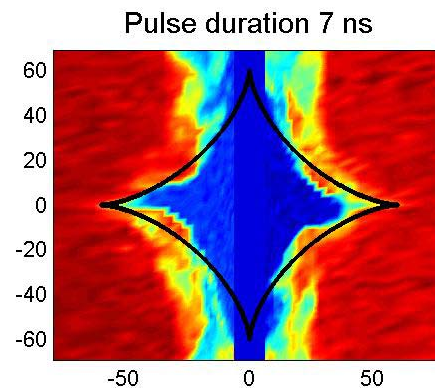
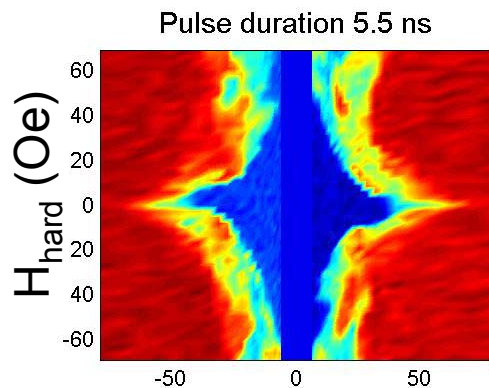
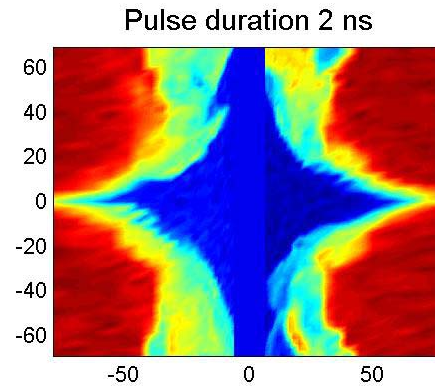
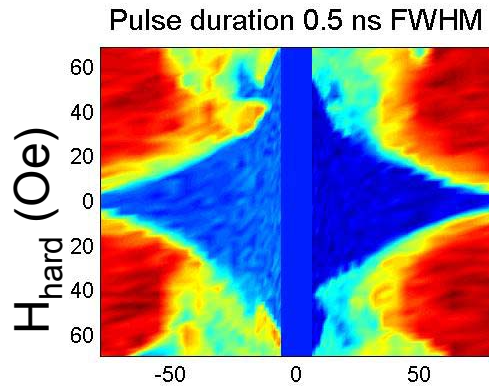
Half – select switching at higher easy-axis field ( $>50$  Oe),  
incomplete switching at lower values.

Normalized easy-axis magnetization, after 0.5 ns (FWHM) pulse

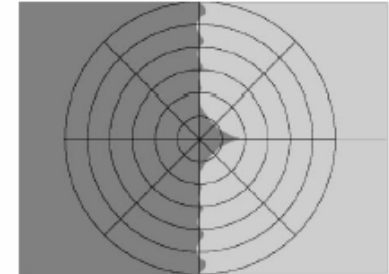




study as a function of pulse durations  
switching diagrams rendered according to final magnetization state



me (pulse length = 1.4 ns)



dynamic switching, final states  
M. Bauer, J. Fassbender, B. Hillebrands,  
and R. L. Stamps, "Switching behaviour  
of a Stoner particle beyond the relaxation  
time limit," Phys. Rev. B **61** (5), 3410-  
3416 (2000)

- switching astroids shrink in the easy axis direction as the pulses duration increase.
- incomplete switching is observed at low easy-axis field values.
- astroids are extended in hard axis direction.
- short pulses (of 0.5 ns FWHM) are sufficient to achieve reliable half-select switching, and their shape doesn't change significantly at pulse durations of more than 5 ns.
- "pockets" of switching are observed at long pulse durations

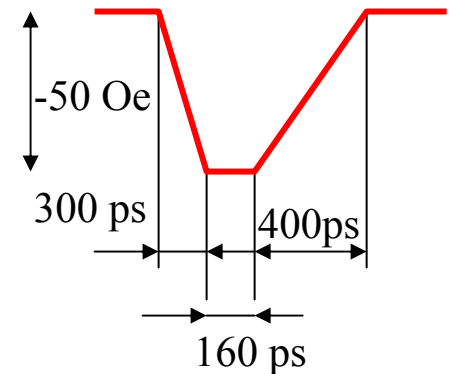
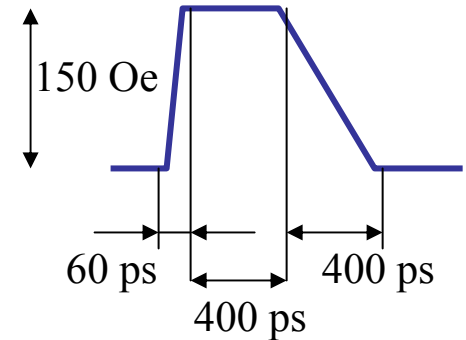
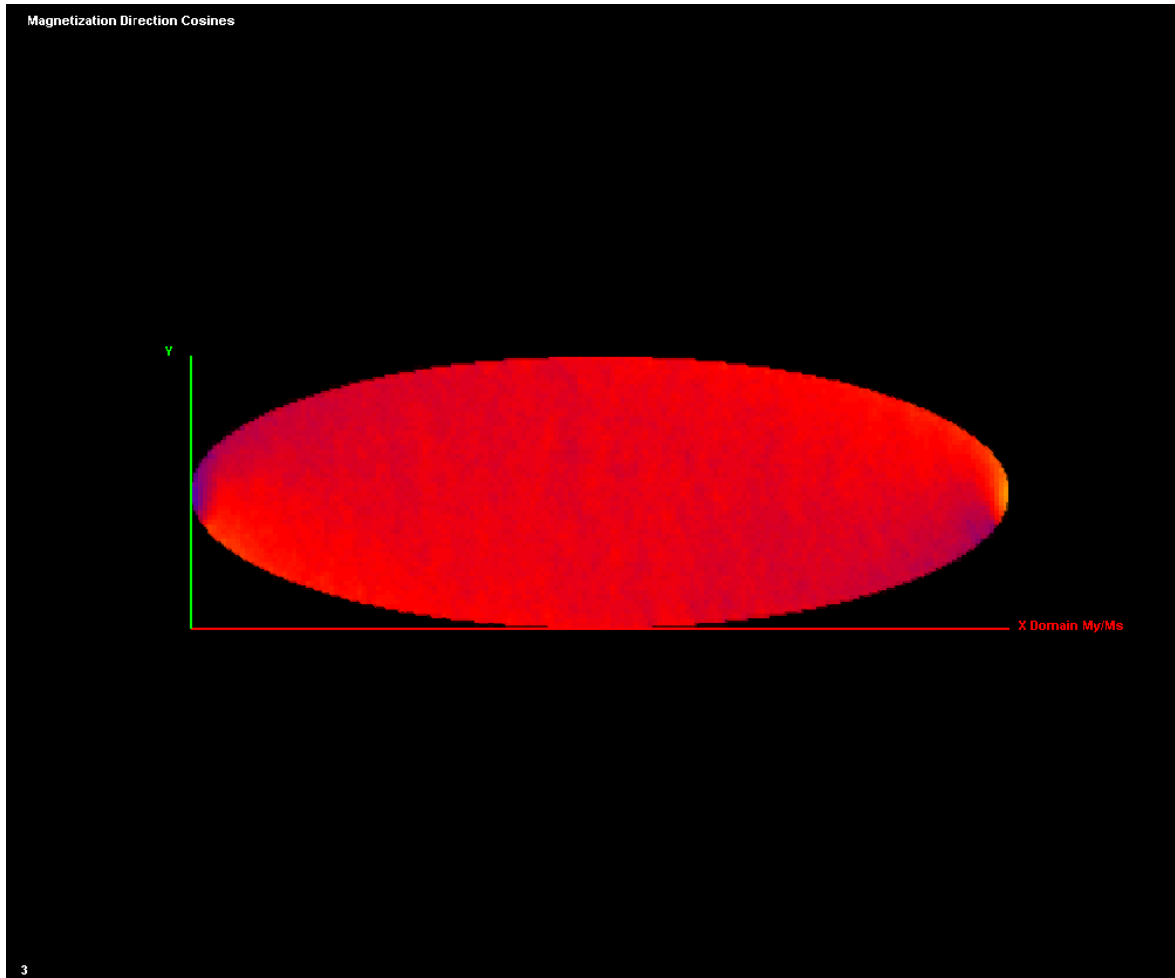


## “incomplete switching”

basic mechanism = vortex formation

A single vortex often is formed in the case of a weak easy-axis field.

Stronger easy-axis field prevents this.



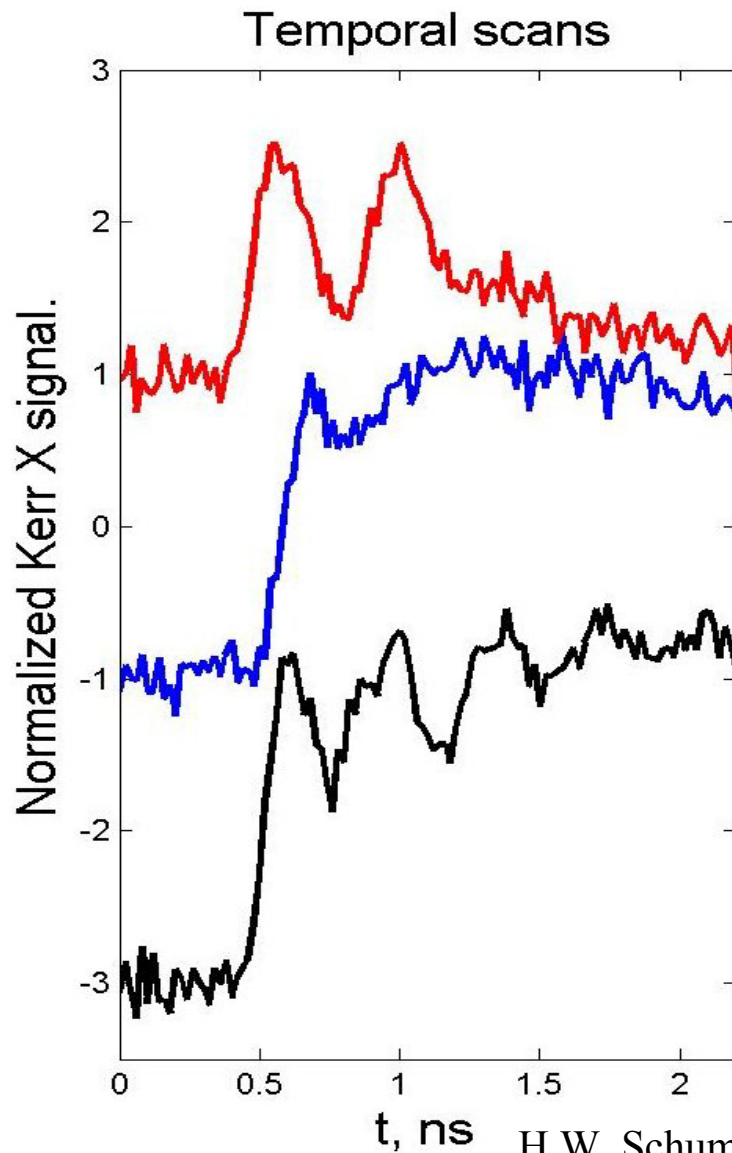
Red = Right

blUe = Up

yellOW = dOWn

grEen = lEft.

# "coherent half-select switching"



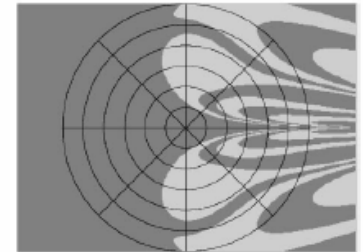
**temporal trace for 220 Oe hard axis field only; unswitched at the end.**

**hard axis pulse assisted by 45 Oe easy-axis field of duration of 0.5 ns Fast switching is achieved, with little ringing.**

**easy axis pulse delayed ~170 ps with respect to the hard-axis pulse. Different phase at end of pulse yields ringing.**



(pulse length = 0.25 ns)



Bauer diagram for SW particle in precessional regime

[other related work:](#)

H.W. Schumacher et. al., Phys. Rev. Lett. **90** (1) 2003 017204-1 – 017204-4

Th. Gerrits. et. al., Nature **418**, 509 (2002).

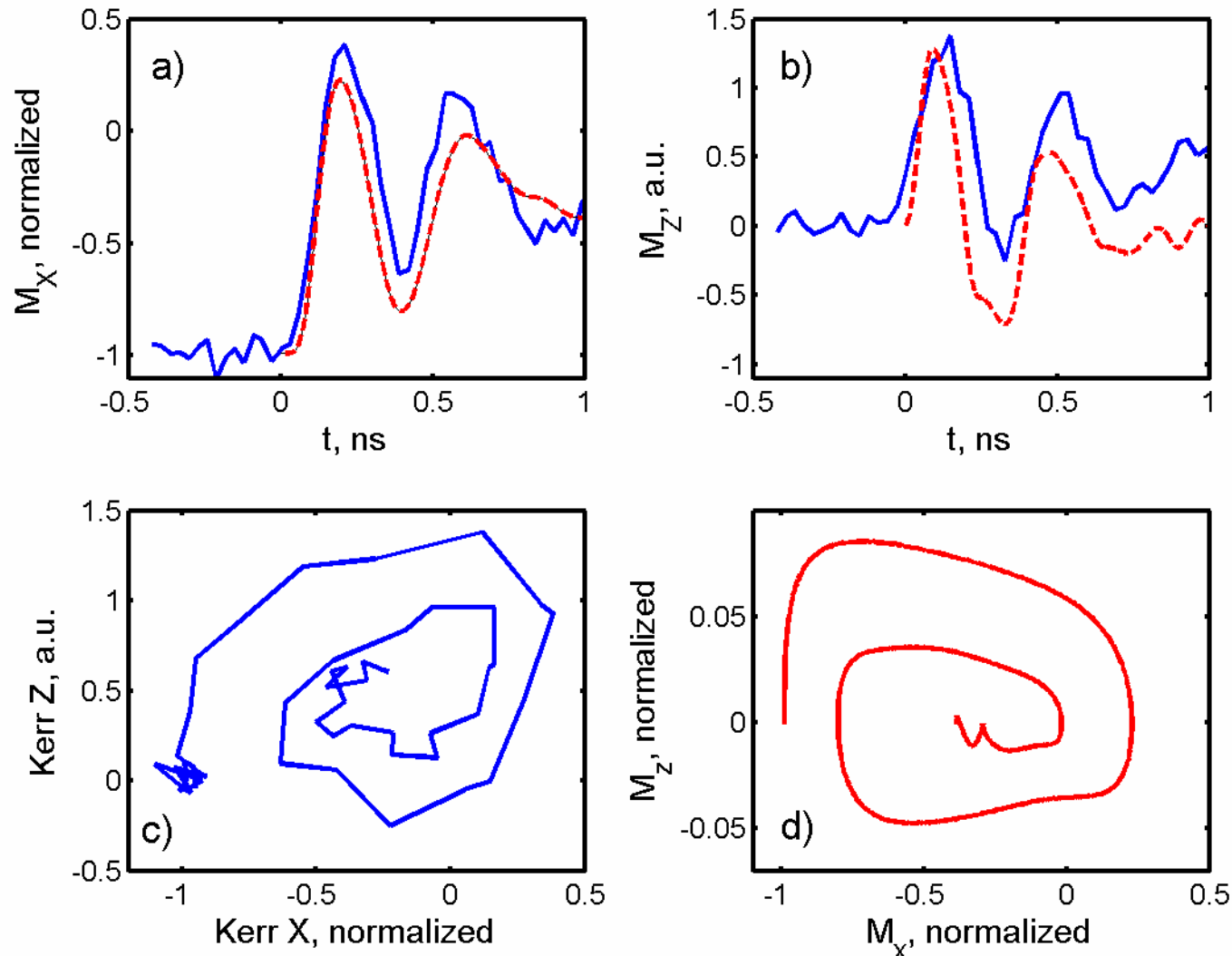
S. Kaka and S.E. Russek, Appl. Phys. Lett. **80**, Feb. 2002.

W.K. Hiebert et. al., J. Appl. Phys. **93**, May 2003.

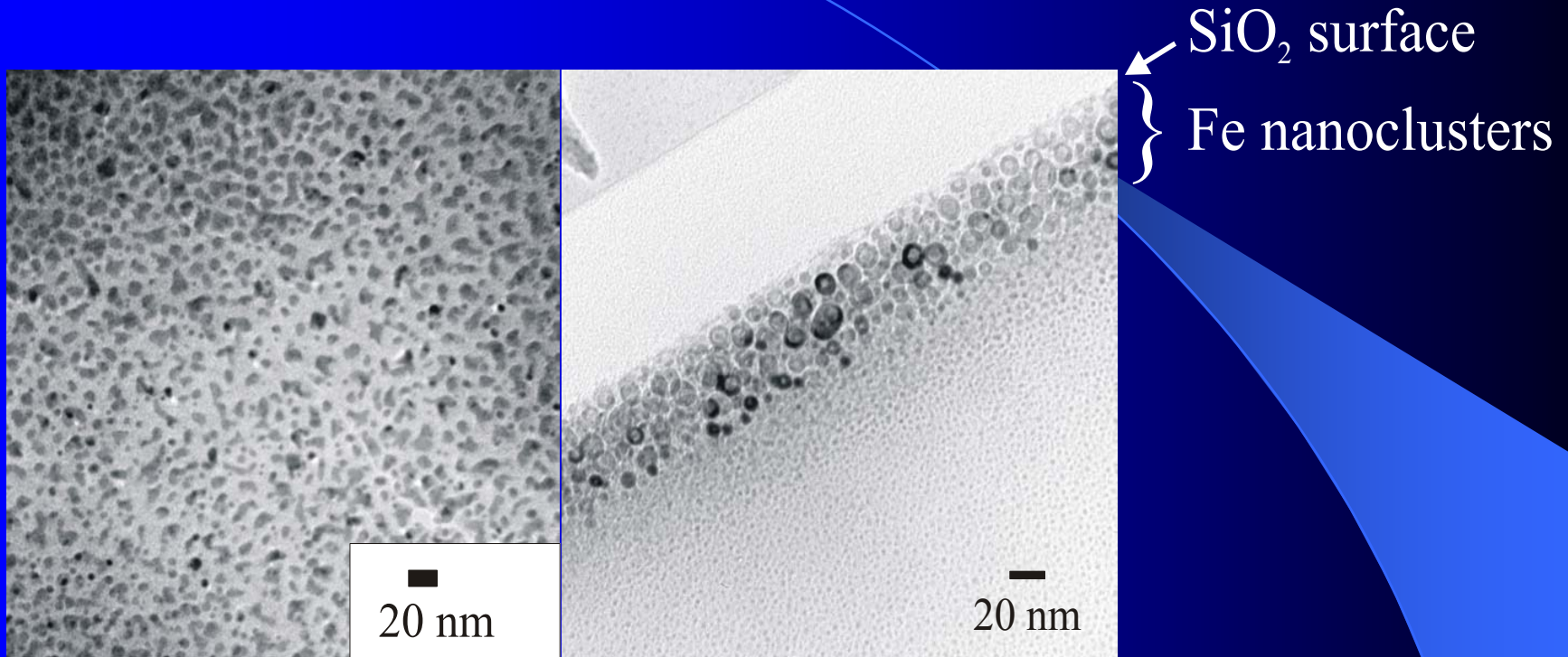
## precessional reversal with fast pulses

experiment vs. simulation,  $H_y = 140$  Oe

Both experimental data and simulated evolution of the magnetization demonstrate similar behavior (spiraling in XZ plane) during the coherent phase of switching.



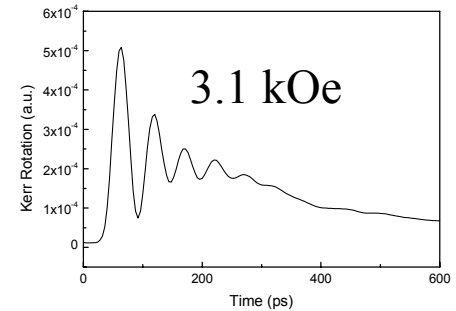
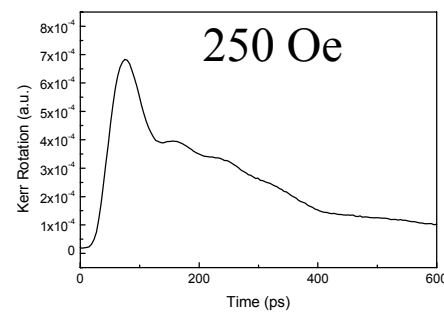
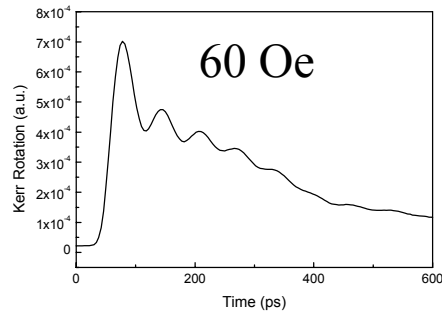
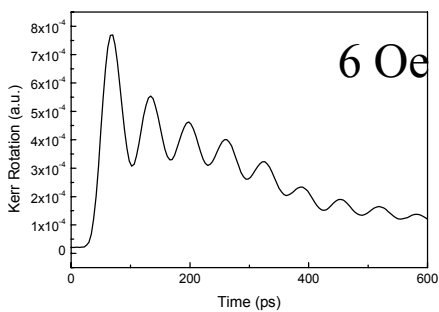
# Iron Nanocrystals in SiO<sub>2</sub>



- Fe ions, 80 keV,  $1.5 \times 10^{17}$  ions/cm<sup>2</sup>
- Amorphous, high purity SiO<sub>2</sub> host
- As implanted

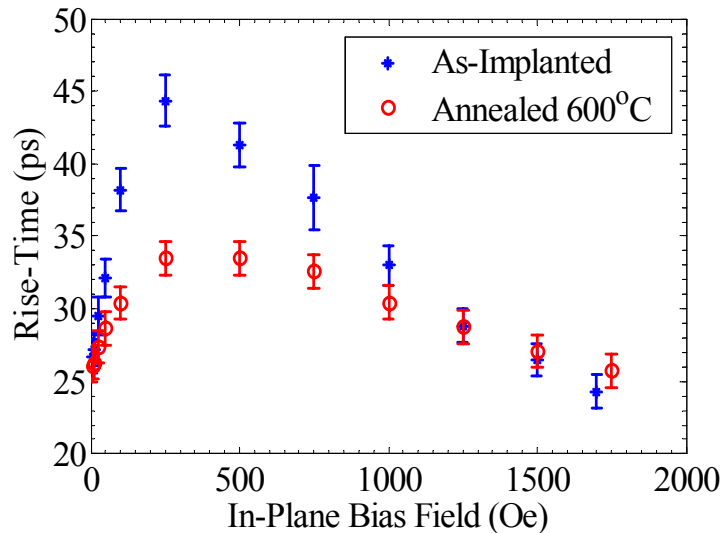
Ion Implantation:  
C.W. White, ORNL

# Iron Implanted (SiO<sub>2</sub>)



Coupling Induced high frequency precession at zero bias

Nonmonotonic change of precessional frequency (inverse rise time) via external bias field



- Zero-field and high-field rise-times fastest
- Mid-field rise-time slowest
  - Slower for as-implanted sample

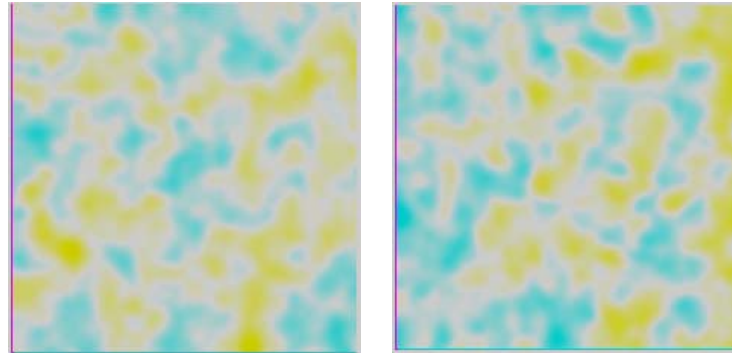
# Experiment & Simulation

Random

Array

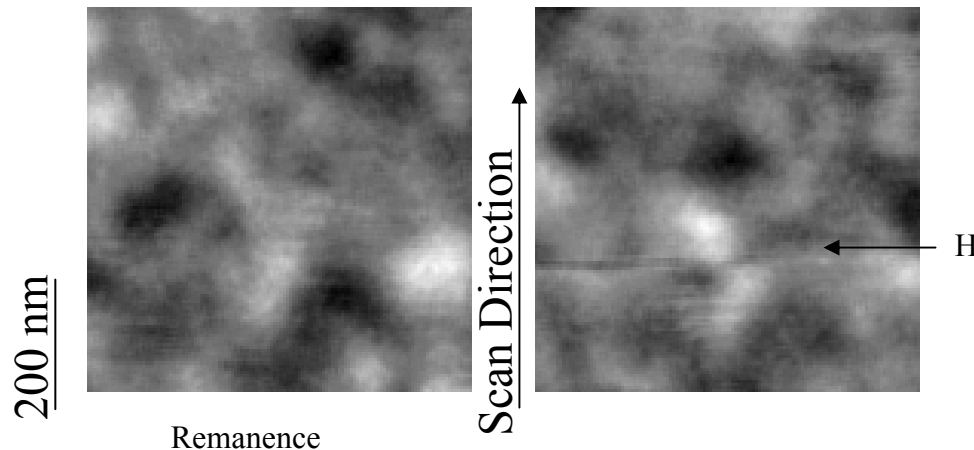
10 nm cubes

2.5 nm cell size

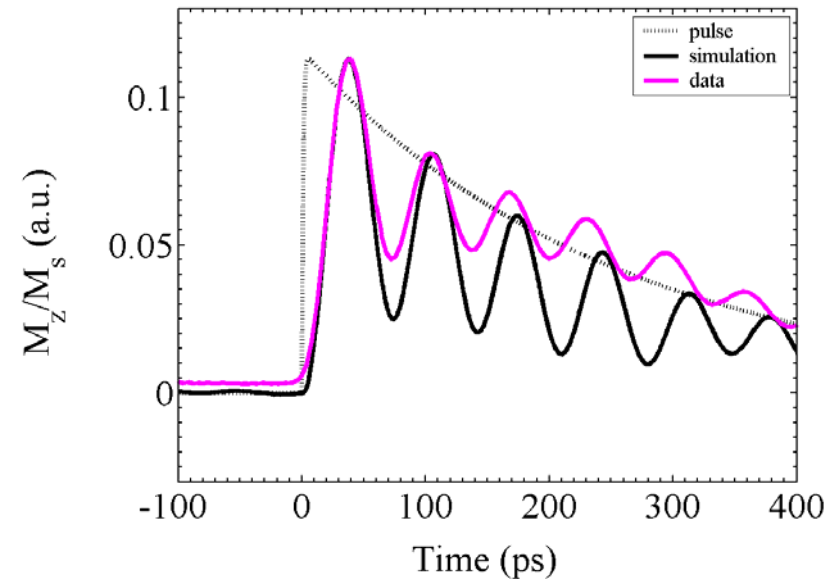


$H_z$  25 nm above  
(simulation)

MFM imaging

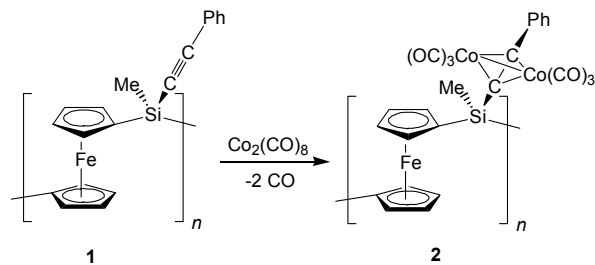


100 Oe





## (ii): nanocomposites created via pyrolysis of organometallic polymer precursor



cobalt-clusterized polyferrocenylsilane (Co-PFS)

Scheme 1

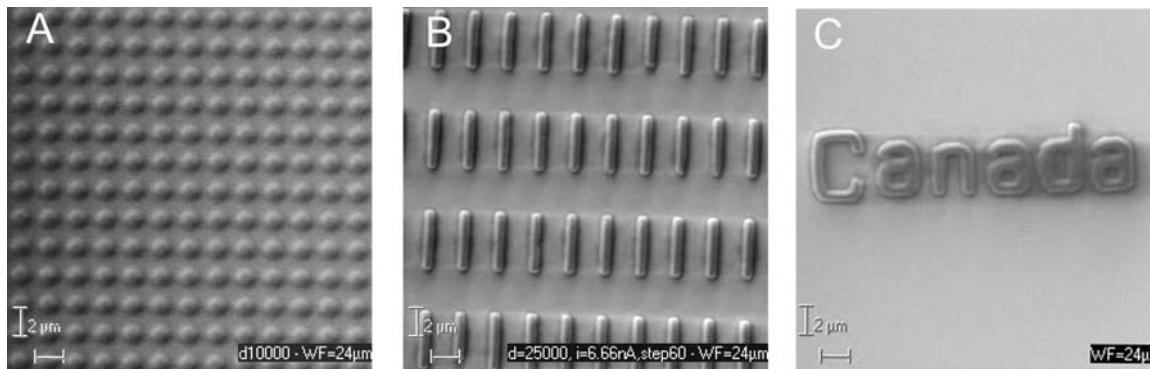


Fig. 1. SEM images of A) dots, B) bars and C) curved lines fashioned by EBL using a Co-PFS resist.

# Summary

Continuous Film



Magnetic Microstructures



Magnetic Nanostructures

- Uniform excitation, control of damping parameter, dynamic coupling (multilayers)
- **Non-uniform magnetization & Pattern Formation**
- **Complex, ultrafast response, coupling**

# prospects

- a grand challenge: 3D spatial imaging. behaviour only uniform through the thickness for quasi-2D samples; many other problems associated with fully exploiting the third dimension, but clearing away one major obstacle would have a big impact.
- devil's advocate question (long-term): synchrotron source vs. ultrafast laser-based x-ray source?